

Specific Heat and Magnetocaloric Effect in Ho-Er-Co Solid Solutions

J. Ćwik · T. Palewski · K. Nenkov

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Abstract Specific heat measurements have been performed on polycrystalline HoCo_2 , ErCo_2 and their solid solutions $\text{Ho}_{1-x}\text{Er}_x\text{Co}_2$ ($0.1 \leq x \leq 0.5$). These compounds were synthesized using high-purity rare-earth metals and cobalt. X-ray diffraction patterns taken at room temperature reveal that all compounds have the C15 cubic Laves phase structure. Heat capacity measurements have been performed in the temperature range of 2–300 K without magnetic field and in a magnetic field of 1 and 2 T. The magnetocaloric effect has been estimated in terms of isothermal magnetic entropy change for all solid solutions in magnetic fields up to 2 T. The effect of increasing Er amount in $\text{Ho}_{1-x}\text{Er}_x\text{Co}_2$ on the magnetic and magnetocaloric properties will be discussed.

Keywords Magnetic properties of magnetically ordered materials (75.30.-m) · Specific heat of magnetic materials (75.40.-s) · Magnetoelectric effects (75.85.+t)

1 Introduction

In recent years, the intermetallic compounds RCo_2 (R = rare earth) belonging to the C15-type cubic Laves-phase crystal structure were the subject of extensive investigations because of their interesting intrinsic magnetic properties. A large magnetocaloric effect (MCE) observed in some of

these compounds makes them suitable candidates for magnetic refrigeration technology [1, 2]. Attractive materials for magnetic refrigeration applications are compounds that display first-order magnetic phase transitions [3]. In this context, RCo_2 intermetallic compounds are one of the promising candidates. It has been found that in the case of RCo_2 with heavy rare-earth elements (R = Ho, Dy and Er) the ferrimagnetic–paramagnetic phase transitions are of first order and these compounds exhibit a large magnetic entropy change [4, 5]. In the RCo_2 compounds the magnetic moment on the Co sites is induced by an exchange interaction with the rare-earth moments [6, 7]. The Co moments range from 0.5 to 0.8 μ_B in the light rare-earth compounds, while they are almost constant ($\sim 1.0 \mu_B$) in the heavy rare-earth compounds. The induced moments are aligned parallel to the 4f moments in the light rare-earth compounds and antiparallel in the heavy rare-earth compounds [8]. From a large number of intermetallic compounds, HoCo_2 and ErCo_2 were selected and their Curie temperature is 75 and 32 K, respectively [9]. It should be noted that MCE properties of HoCo_2 and ErCo_2 were investigated. According to Tishin et al. [10] for HoCo_2 the maximum ΔT_{ad} at 84 K reaches 10 K for a field change $\mu_0 H$ of 10 T. Giguere et al. [11] reported the maximum $\Delta T_{ad} = 14.4$ K for ErCo_2 at 33 K for $\mu_0 H = 14$ T. In the case of $\text{Ho}_{1-x}\text{Er}_x\text{Co}_2$ some theoretical calculation was performed by Oliveira et al. [12]. They have shown that the introduction of Er ions decreases the magnetic ordering temperature of the pseudobinary compounds and the nature of the magnetic phase transitions continues to be first-order for all Er concentrations.

In the present work, the influence of the partial replacement of Ho by Er in $\text{Ho}_{1-x}\text{Er}_x\text{Co}_2$ solid solutions on the phase structure and MCE has been studied experimentally.

J. Ćwik (✉) · T. Palewski · K. Nenkov
International Laboratory of High Magnetic Fields and Low
Temperatures, Gajowicka 95, 53-421 Wrocław, Poland
e-mail: cwikjac@ml.pan.wroc.pl

K. Nenkov
IFW Dresden, Postfach 270016, 01171 Dresden, Germany

2 Experimental Details

Polycrystalline samples of $\text{Ho}_{1-x}\text{Er}_x\text{Co}_2$ ($0.1 \leq x \leq 0.5$) solid solutions were prepared by arc-melting stoichiometric proportions of starting materials (of at least 99.9 % purity) on a water-cooled copper crucible under a high-purity argon atmosphere. The alloys were remelted four times to ensure good homogeneity. The mass losses after the melting were less than 1 wt%. The buttons obtained were wrapped in a tantalum foil, sealed in evacuated quartz ampules, and annealed at 700 °C for two weeks.

Specific heat measurements were performed in the temperature range of 2–300 K in applied magnetic fields of 1 and 2 T using Quantum Design PPMS 14 Heat Capacity System. X ray powder diffraction revealed that $\text{Ho}_{1-x}\text{Er}_x\text{Co}_2$ samples are single phase and have the cubic C15 structure (space group Fd-3m). The lattice parameters decrease with increasing the Er content from 0.7171 for $x = 0.1$ to 0.7164 for $x = 0.5$.

3 Results and Discussion

The specific heat of $\text{Ho}_{1-x}\text{Er}_x\text{Co}_2$ measured in zero magnetic field are shown as functions of temperature in Fig. 1. It can be seen that sharp peaks associated with the first-order magnetic phase transitions were observed. The Curie temperatures decrease from 73 K in $\text{Ho}_{0.9}\text{Er}_{0.1}\text{Co}_2$ down to 50 K in $\text{Ho}_{0.5}\text{Er}_{0.5}\text{Co}_2$. Specific heat measurements in a magnetic field show that the maximum $C_p(T)$ peaks shifted to higher temperatures, and were gradually decreases with increasing magnetic field (see inset of Fig. 1). Moreover for the samples with $x \leq 0.4$ additional relatively small and smooth step-like increase in $C_p(T)$ plots at the low-temperature region is observed. One possible mechanism

for the low-temperature transition might be a spin reorientation in the Ho(Er) 4f subsystem similar to that observed in HoCo_2 ($T_{\text{SR}} = 15$ K) and NdCo_2 ($T_{\text{SR}} = 43$ K) [13]. In those two compounds it is assumed that the spin reorientation is driven by an interplay of molecular and crystal fields, thus giving rise to the change of the easy direction of magnetization from [1 0 0] to [1 1 0] with decreasing temperature. The substitution of Er for Ho in $\text{Ho}_{1-x}\text{Er}_x\text{Co}_2$ results in the increase in the spin-reorientation temperatures from 24 for $x = 0.1$ to 45 for $x = 0.4$. The spin-reorientation transition, observed in $\text{Ho}_{1-x}\text{Er}_x\text{Co}_2$ with a maximal 40 % substitution of Er for Ho can be understood as an effect of the competition between the magnetocrystalline anisotropy of the Co and Ho(Er) sublattices.

The relatively low Curie temperature and spin-reorientation effect does not allow for an accurate determination of the electronic heat capacity coefficient γ and Debye temperature Θ_D in the low-temperature range. In order to estimate the magnetic contribution a theoretical calculation of the Debye function was made. The fitting was performed according to the formula [14]:

$$C_{\text{el+ph}}(T) = \gamma T + 9NR \left(\frac{T}{\Theta_D} \right)^3 \int_0^{\Theta_D/T} \frac{x^4 e^x}{(e^x - 1)^2} dx, \quad (1)$$

where the first term represents the electronic heat capacity and the second term corresponds to the phonon Debye contribution; $N = 3$ is the number of atoms per formula unit; R is the universal gas constant; and $x \equiv \hbar\omega/k_B T$.

The best fit for the wide temperature range could be obtained by fixing the parameters $\gamma = 30$ mJ/mol K² for all the measured samples, while the Debye temperature increases from 240 to 280 K with the increase of Er content.

The magnetic part of the entropy S_{mag} was calculated by integrating $C_{\text{mag}}(T)/T$. Figure 2 shows, as example, the temperature variation of the magnetic entropy in zero,

Fig. 1 The specific heat $C_p(T)$ of $\text{Ho}_{1-x}\text{Er}_x\text{Co}_2$ in a zero magnetic field. The insets shows the $C_p(T)$ curve in 0, 1 and 2 T magnetic field for $x = 0.1$ near its ordering temperature

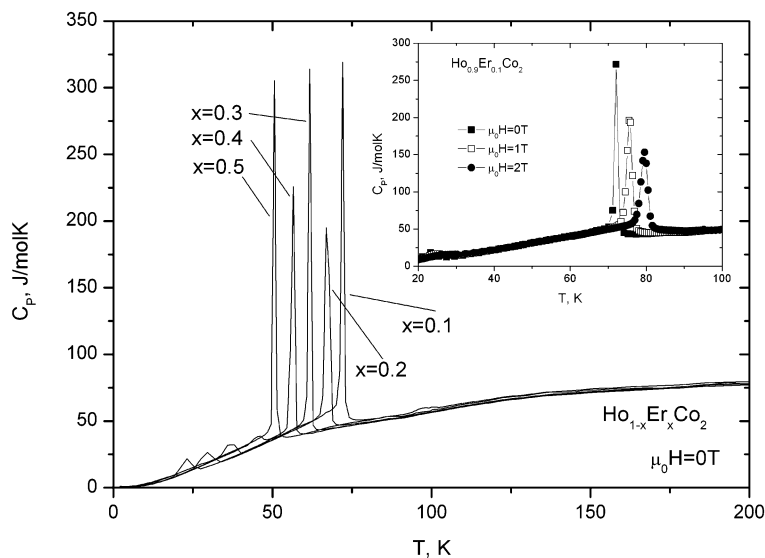


Fig. 2 Magnetic part of the entropy S_{mag} of $\text{Ho}_{0.9}\text{Er}_{0.1}\text{Co}_2$ measured in zero, 1 and 2 T magnetic fields. The insets show a low-temperature portion

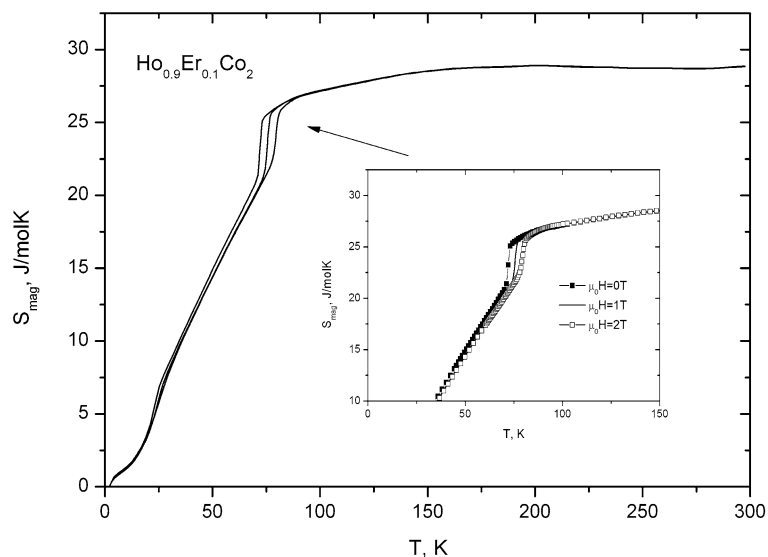
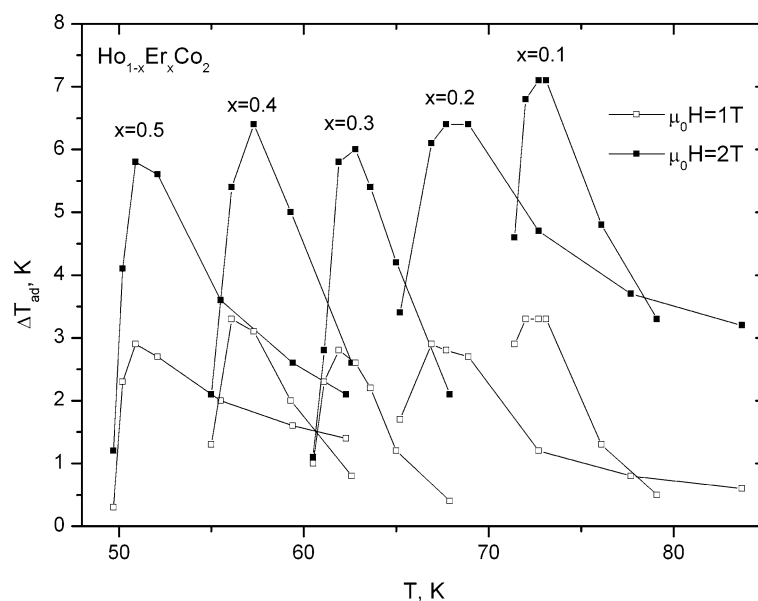


Fig. 3 Temperature dependence of the adiabatic temperature change ΔT_{ad} of $\text{Ho}_{1-x}\text{Er}_x\text{Co}_2$ ($x = 0.1, 0.2, 0.3, 0.4, 0.5$) solid solutions near its ordering temperature under 1 and 2 T magnetic field



1 and 2 T magnetic fields for $\text{Ho}_{0.9}\text{Er}_{0.1}\text{Co}_2$. S_{mag} increases with increasing temperature and, as expected, the application of the magnetic field leads to the decrease of S_{mag} in the vicinity of T_C . For $\text{Ho}_{0.9}\text{Er}_{0.1}\text{Co}_2$ the magnetic entropy reach the maximum values 26.5 J/mol K in the temperature ~ 200 K. In the case of parent compounds it was found that experimental values of the magnetic entropy of HoCo_2 and ErCo_2 at 200 K are 24 and 23.1 J/mol K, respectively [6, 15]. This fact indicates that the magnetic entropy is determined by Ho(Er) moments and that the Co moments make a small contribution to the total S_{mag} .

The magnetocaloric effect was calculated by a method suggested by von Ranke et al. [16]. Figure 3 shows the temperature dependence the adiabatic temperature change $\Delta T_{\text{ad}}(T)$ for all measured solid solutions at a magnetic field of 1 and 2 T. The maximum peak value decreases with in-

creasing Er content. As expected, the magnetic field leads to an increase of ΔT_{ad} . The maximum ΔT_{ad} reaches about 7 K near 73 K for $\text{Ho}_{0.9}\text{Er}_{0.1}\text{Co}_2$ and decreases to about 6 K near 50 K for $\text{Ho}_{0.5}\text{Er}_{0.5}\text{Co}_2$ (magnetic field of 2 T). The sharp peaks in the ΔT_{ad} curves are related to the ferrimagnetic–paramagnetic phase transition at the Curie point, where the magnetization change is maximal.

4 Conclusion

The effect of the partial replacement of Ho by Er in $\text{Ho}_{1-x}\text{Er}_x\text{Co}_2$ solid solutions on the structure and physical properties has been studied. It was confirmed that the main phase in all solid solutions is cubic C15 structure. The substitution of Er in $\text{Ho}_{1-x}\text{Er}_x\text{Co}_2$ results in the increase in

the temperatures T_{SR} and decrease in T_C , as observed in the specific heat measurements. The T_{SR} temperature increases from 24 K to 45 K, and T_C decreases from 73 K to 50 K. The maximum MCE estimated in terms of isothermal magnetic entropy change reaches 7 K (magnetic field of 2 T) for $x = 0.1$ and is highly attractive for magnetic cooling applications.

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